

FFT-BASED SOLVERS TO EVALUATE STRESS DISTRIBUTIONS IN RPV STEELS

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CEA Paris-Saclay DEN/DMN/SRMA

5th September 12h20-13h



□ GENERAL Context

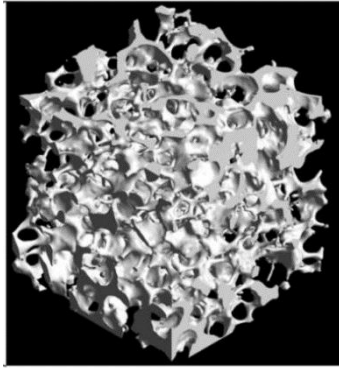
- FFT based solvers for heterogeneous materials
- The AMITEX_FFTP code (specificities and use)

□ SOTERIA Context

- Stresses at Grain Boundaries
- Application to RPV steels

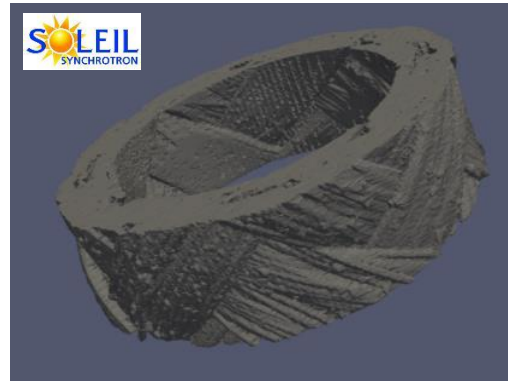
Heterogeneous materials

- Porous ceramics



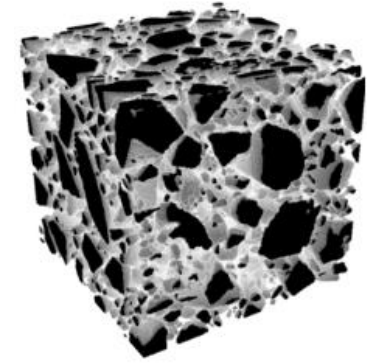
Ackermann & al. Materials 2014

- SiC/SiC composite tube



from CHEN Y. Thesis, CEA, ENPC, 2017

- Concrete



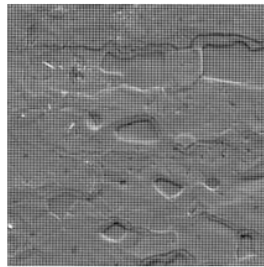
from F. Bernachy, CEA, 2017

- Polycrystals => SOTERIA application!

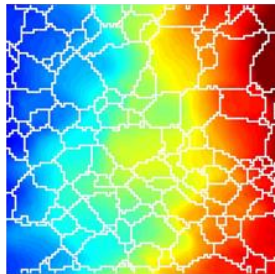
50 μ m



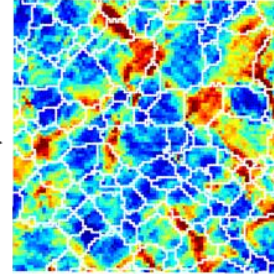
M. Dexet thesis, CEA, LMS-X, 2006



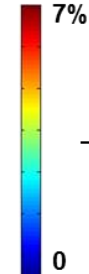
Microgrid



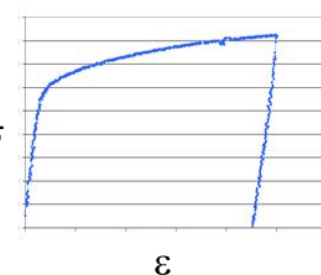
Experimental displacement field



Experimental strain field



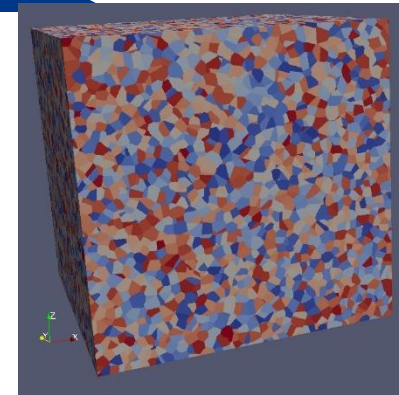
+ σ



Macroscopic response

□ Simulation of heterogeneous materials

- A « **Representative** » **Volume Element (RVE)**
- A **Constitutive behavior law** for each phase
- An « **Average loading** » : uniaxial stress (tensile test) for example
- A type of **Boundary Conditions: Periodic BC** is a good choice



□ Natural trends

- **Increase the spatial resolution** to obtain a better description of local fields
- **Increase the size of the RVEs** to obtain representative results
- « **Physically based** » constitutive behaviors are more and more complex

□ Standard FEM solvers

- **Numerical limits (memory size & computation time)**



□ « FFT-based » solvers

- **No tedious meshing procedure (input=digital image)**
- **Much more efficient than standard FEM solvers**
- **Easy to implement**
- **Well-suited for Parallelism => PUSH BACK THE LIMITS!**



FFT-based solvers for heterogeneous materials



□ FIX-POINT algorithm (Moulinec-Suquet 1994)

Problem to solve

Auxiliary problem

$$\sigma(x) = c(x) : \varepsilon(x)$$

$$\text{div}(\sigma(x)) = 0$$

$$\langle \varepsilon(u(x)) \rangle = E$$

+ periodicity + compatibility

$$\sigma(x) = c_0 : \varepsilon(x) + \tau(x)$$

$$\text{div}(\sigma(x)) = 0$$

$$\langle \varepsilon(u(x)) \rangle = E$$

+ periodicity + compatibility

$$\sigma(x) = c_0 : \varepsilon(x) + \underbrace{(c(x) - c_0) : \varepsilon(x)}_{\tau(x)}$$



Rewriting of the problem

$$\sigma(x) = c_0 : \varepsilon(x) + \tau(x)$$

$$\tau(x) = (c(x) - c_0) : \varepsilon(x)$$

$$\text{div}(\sigma(x)) = 0$$

$$\langle \varepsilon(u(x)) \rangle = E$$

+ periodicity + compatibility

Solution for the auxiliary problem

$$\varepsilon(x) = -\Gamma_0 * \tau(x) + E$$

Applying the Green operator
Simple in Fourier space (FFT)
Mura 1997

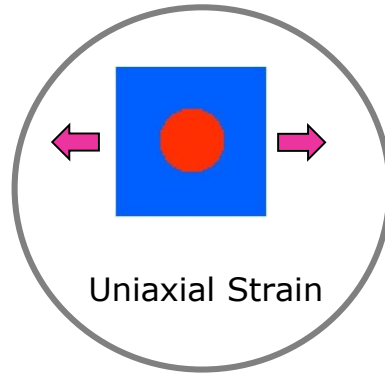
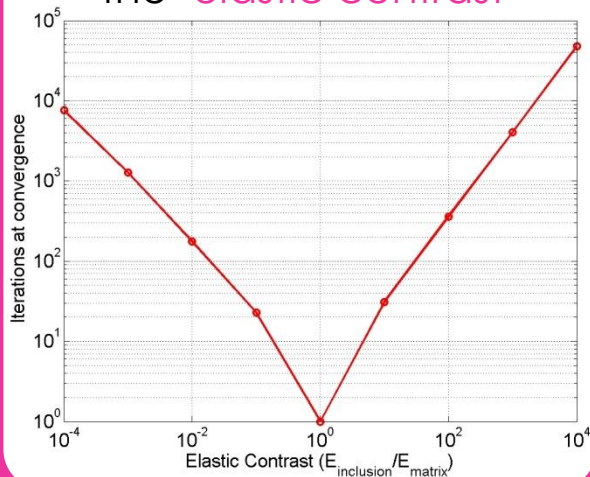
Moulinec-Suquet, 1994

$$\tau(x) = (c(x) - c_0) : \varepsilon(x)$$

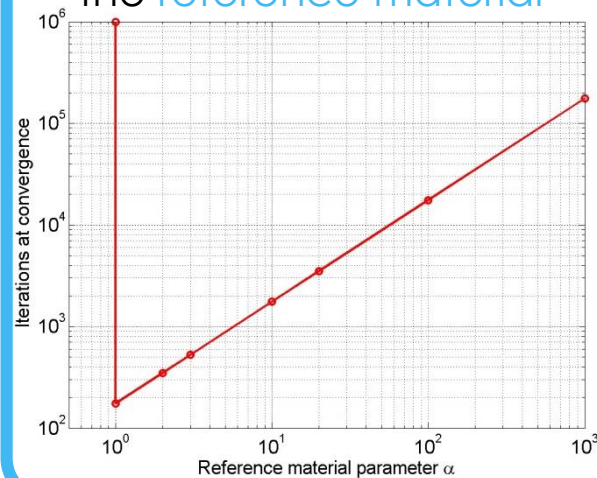
FFT-based solvers for heterogeneous materials

- Drawbacks of the method as proposed in 1994 by Moulinec & Suquet

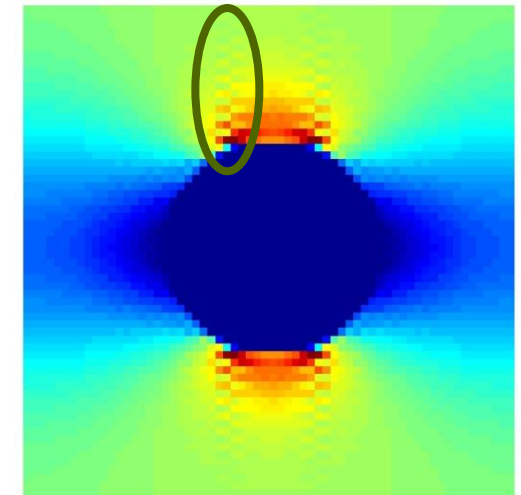
Sensitivity to the elastic contrast



Sensitivity to the reference material



Spurious Oscillations



- A lot of work and improvements since 1994!

- **Modified Discrete Green Operator** (*Willot 2015, Schneider 2016...*)

- sensitivity to the **elastic contrast**
- **spurious oscillations**

- **Algorithms** (*Zeman 2010, Brisard 2010, Gélébart 2013...*)

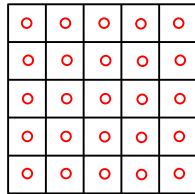
- sensitivity to the **reference material**

- **Composite voxels** (*Kabel2015, Gélébart 2015...*)

- **spurious oscillations**
(very efficient for thin interphases)

□ Discrete Green operators

- **CLASSICAL DGO : truncated Continuous Green Operator** (Moulinec-Suquet 1994)



$$\begin{aligned} \mathbf{div}(\boldsymbol{\sigma}) &= \mathbf{0} \\ \boldsymbol{\varepsilon} &= (\mathbf{grad}(\mathbf{u}))^{sym} \\ \boldsymbol{\sigma} &= \mathbf{c}_0 : \boldsymbol{\varepsilon} + \boldsymbol{\tau} \end{aligned}$$



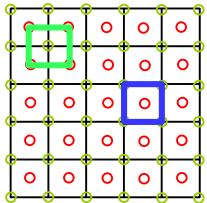
$$\begin{aligned} i\hat{\sigma} \cdot \mathbf{k}_a &= \mathbf{0} \\ \hat{\boldsymbol{\varepsilon}} &= i\widehat{\mathbf{u}}^* \otimes^{sym} \mathbf{k}_a \\ \hat{\sigma} &= \mathbf{c}_0 : \hat{\boldsymbol{\varepsilon}} + \hat{\boldsymbol{\tau}} \end{aligned}$$



$$\hat{\boldsymbol{\varepsilon}} = -\widehat{\Gamma}_0 : \hat{\boldsymbol{\tau}}$$

$\mathbf{u}, \boldsymbol{\varepsilon}, \boldsymbol{\sigma}$ at voxels centers

- **MODIFIED DGO : DISCRETE DIFFERENTIAL OPERATORS = contour integrals**



\mathbf{u} at voxels **corners** \Rightarrow $\boldsymbol{\varepsilon} = (\mathbf{grad}(\mathbf{u}))^{sym} \cong \frac{1}{v} \int_{\partial v} \mathbf{u} \otimes^{sym} \mathbf{n} ds$ at voxels **centers**

$\boldsymbol{\sigma}$ at voxels **centers** \Rightarrow $\mathbf{div}(\boldsymbol{\sigma}) \cong \frac{1}{v} \int_{\partial v} \boldsymbol{\sigma} \cdot \mathbf{n} ds$ at voxels **corners**



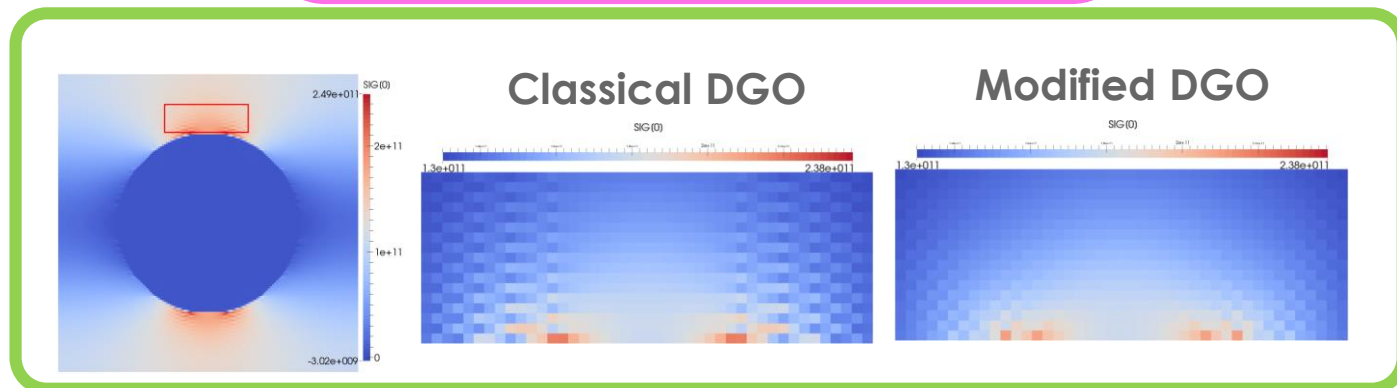
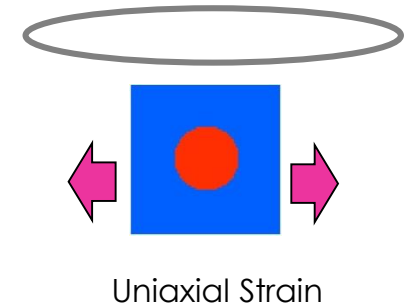
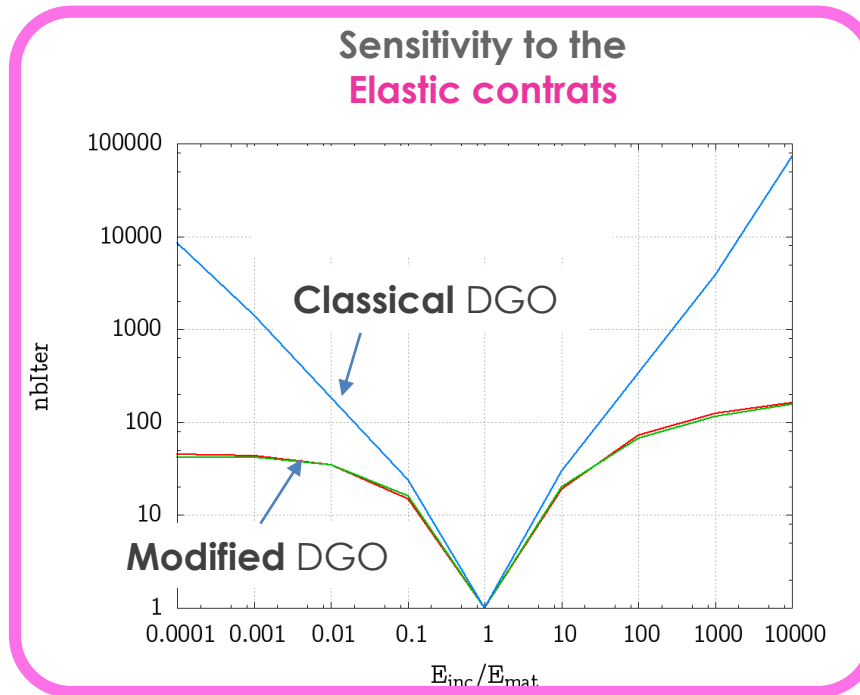
$$\begin{aligned} i\hat{\sigma} \cdot \widetilde{\mathbf{k}}_a &= \mathbf{0} \\ \hat{\boldsymbol{\varepsilon}} &= i\widehat{\mathbf{u}}^* \otimes^{sym} \widetilde{\mathbf{k}}_a \\ \hat{\sigma} &= \mathbf{c}_0 : \hat{\boldsymbol{\varepsilon}} + \hat{\boldsymbol{\tau}} \end{aligned}$$



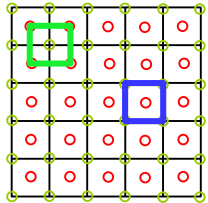
$$\hat{\boldsymbol{\varepsilon}} = -\widehat{\Gamma}_0 : \hat{\boldsymbol{\tau}}$$

FFT-based solvers for heterogeneous materials

Modified Discrete Green Operator

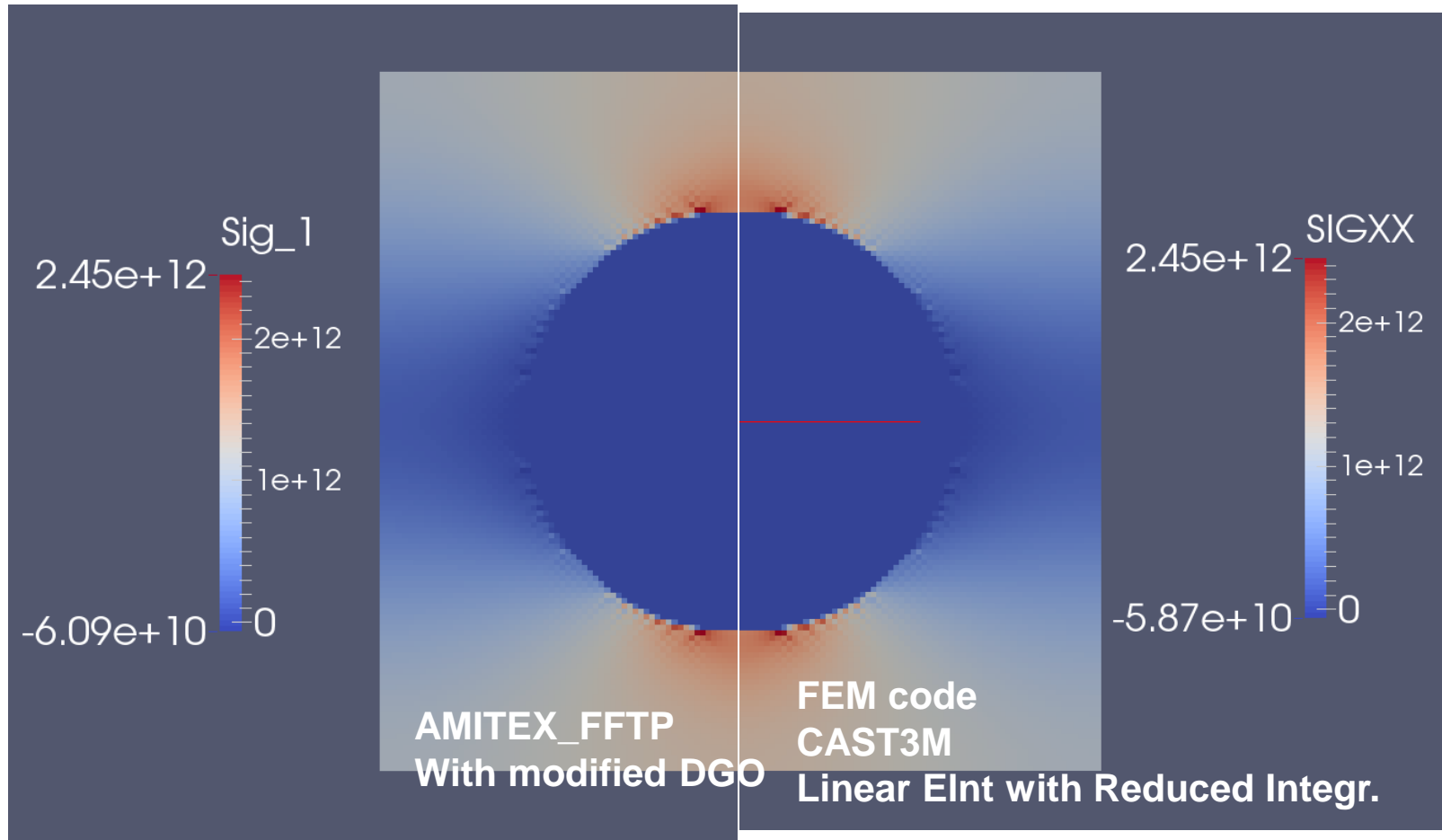


□ Modified Discrete Green Operator

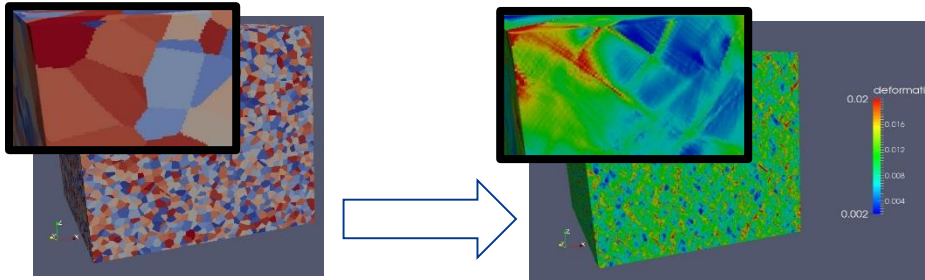


Linear FINITE ELEMENTS with Reduced integration (Schneider & al., IJNME 2016)

□ Modified Discrete Green Operator



➔ AMITEX_FFTP : a **FE method** with an **FFT-based solver**



AMITEX

<http://www.maisondelasimulation.fr/projects/मितex/html/overview.html>

old version on the website, to be refreshed
contact L. Gelebart for a recent version

□ Highly parallel implementation (MPI)

▪ Models

- Mechanics : **SMALL STRAINS** and **FINITE STRAINS**
- Diffusion

▪ Algorithm

- Fix Point + Convergence acceleration

▪ Behavior

- User defined : **umat** compatibility => **coupling with mfront!**
- « Composite » voxels

▪ Various loading types

The AMITEX_FFTP code



- Highly parallel implementation (distributed memory with MPI)

$$\varepsilon^0(x) = E$$

$$\tau^k(x) = \sigma(\varepsilon^k(x)) - c_0 : \varepsilon^k(x)$$

$$\tau^k(x) \rightarrow \hat{\tau}^k(\xi)$$

$$\hat{\varepsilon}^{k+1}(\xi) = -\hat{\Gamma}_0(\xi) : \hat{\tau}^k(\xi) \quad \hat{\varepsilon}^{k+1}(0) = E$$

$$\hat{\varepsilon}^{k+1}(\xi) \rightarrow \varepsilon^{k+1}(x)$$

Distributed memory //
implementation

(MPI)



- Behavior : « local » in real space

- Green Operator : « local » in Fourier space

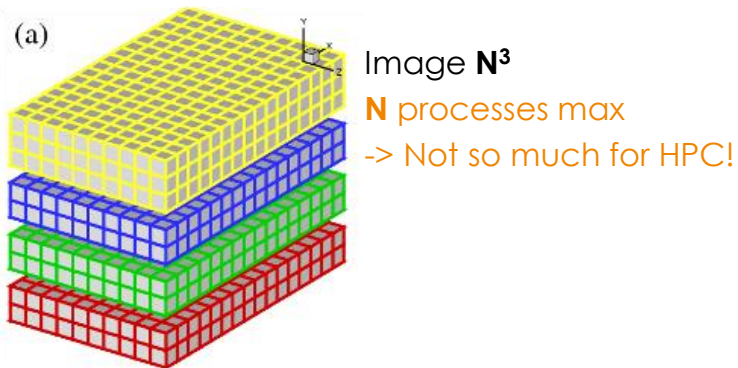
- FFT & iFFT : « non-local » (needs data transfer)

The AMITEX_FFTP code

□ Highly parallel implementation (MPI)

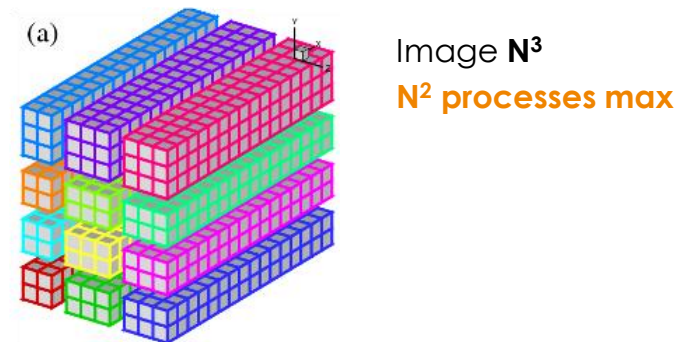
▪ Decomposition

✓ Decomposition 1D (slabs)



✓ Decomposition 2D (pencils)

<http://www.2decomp.org/>

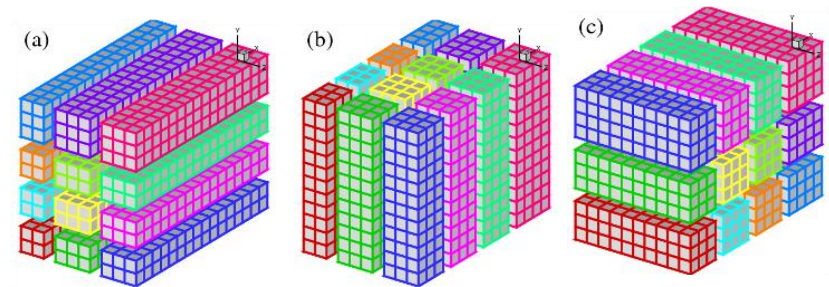


<http://www.2decomp.org/>

▪ 3D-FFT = succession of 1D-FFT

Requires the transposition of data

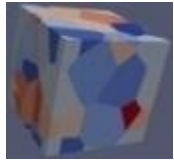
- Communications (MPI_ALLTOALL!)
 - 2decomp library



The AMITEX_FFTP code



□ Highly parallel implementation (MPI)



- Polycrystal (voronoï), **dislocation-based Crystal Plasticity (49 var.int.)**, Small Strains
- Cluster poincare (Maison de la Simulation) 16 cores (2x8) / node sandy bridge E5-2670

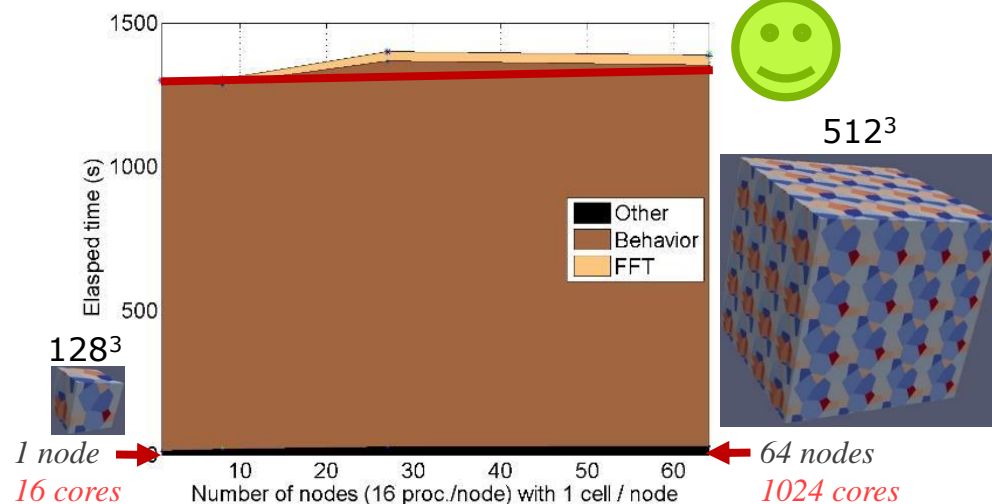
Weak scalability

Number of nodes = **N**, Problem size = **NxK0**

Elapsed time on 1 node : t_{ref}

Elapsed time on N nodes : t_N

IDEALLY : $t_N = t_{ref}$



The AMITEX_FFTP code



```
mpirun amitex_fftp -nm mate.vtk -nz zone.vtk -m Material.xml -c Loading.xml -a Algorithm.xml -s result
```



To run amitex in //

Geometry

The output

□ Input geometry : 3D images (vtk format) : **mate.vtk** and **zone.vtk**

The microstructure consists of :

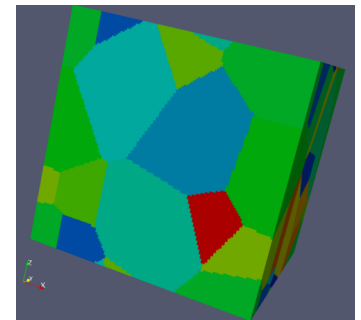
- one or different **materials** (one material = one constitutive law)
mate.vtk : 3D image defining the ID of the materials
- each material can be divided in different **zones** (where the coefficients are constants)
zone.vtk : 3D image defining the ID of the zones

In our case (a polycrystal) :

- only ONE material : “-nm **mate.vtk**” can be omitted (the ID is 1 everywhere)
- the grain definition is given by the 3D image **zone.vtk**

□ The default output :

- unit cell and “per material” average (and std dev.) of stresses and strains at each computation time



□ Algorithm.xml

Algorithm = **Fix-point** (Basic_Scheme)
+ **Convergence acceleration**
+ default criterion : 10^{-4}

Use of the **modified Discrete Green Operator** (*Filter*
Type="Default")
+ **Finite Strains** framework

```
<?xml version="1.0" encoding="UTF-8"?>
<Algorithm_Parameters>
  <Algorithm Type="Basic_Scheme">
    <Convergence_Acceleration Value="true"/>
    <Convergence_Criterion Value="Default"/>
  </Algorithm>
  <Mechanics>
    <Filter Type="Default"/>
    <Small_Perturbations Value="false"/>
  </Mechanics>
</Algorithm_Parameters>
```

□ Loading.xml

Additional outputs :

Stress and strain fields
« Per zone » average and std dev.

Computation time discretization

Increments for fields outputs (vtk format)
Number of « per zone » output (equally spaced in time)

Evolution the average applied tensor :
9 components
if Driving = Stress : Piola-Kirchoff stress
if Driving = Strain : Displacement
gradient

Remark : 6 components in small strain framework
if Driving = Stress : Cauchy stress
if Driving = Strain : Linearized Strain

```
<?xml version="1.0" encoding="UTF-8"?>
<Loading_Output>

<!-- OUPUT QUANTITIES -->
<Output>
  <vtk_StressStrain Strain = "1" Stress = "1"/>
  <Zone numM="1"> </Zone>
</Output>

<!-- SUCCESSIVE LOADINGS (only one here)-->
<Loading Tag="1">
  <Time_Discretization Discretization="Linear" Nincr="1000" Tfinal="333.33"/>

  <Output_vtkList>18 40 80 160 320 640 1000</Output_vtkList>
  <Output_zone Number="50"/>

  <!-- tensile test in the z direction -->
  <xx Driving="Stress" Evolution="Constant" />
  <yy Driving="Stress" Evolution="Constant" />
  <zz Driving="Strain" Evolution="Linear" Value="0.1" />
  <xy Driving="Stress" Evolution="Constant" />
  <xz Driving="Stress" Evolution="Constant" />
  <yz Driving="Stress" Evolution="Constant" />
  <yx Driving="Strain" Evolution="Constant" />
  <zx Driving="Strain" Evolution="Constant" />
  <zy Driving="Strain" Evolution="Constant" />
</Loading>

</Loading_Output>
```

The AMITEX_FFTP code



□ Material.xml

Lamé coefficient of the reference material

Name of the dynamic library + Name of the behavior law

Constant coefficients

Constant per zone coefficients :
=> Euler angles of each grain

Initial internal variables

```
<?xml version="1.0" encoding="UTF-8"?>
<Materials>
  <!-- REFERENCE MATERIAL -->
  <Reference_Material Lambda0=" 2.0431e+5" Mu0="0.8756e+5"/>

  <!-- MATERIAL 1 -->
  <Material numM="1" Lib="/home/gelebart/SIMULATIONS/libUmatAmitex.so" Law="umatBCCSOTERIA" >

    <Coeff Index="1" Type="Constant" Value="236.412E3"/>
    .
    .
    <Coeff Index="27" Type="Constant" Value="200"/>
    <Coeff Index="28" Type="Constant_Zone" File="/home/gelebart/SIMULATIONS/PHI1.or" />
    <Coeff Index="29" Type="Constant_Zone" File="/home/gelebart/SIMULATIONS/PHI.or" />
    <Coeff Index="30" Type="Constant_Zone" File="/home/gelebart/SIMULATIONS/PHI2.or" />

    <IntVar Index="1" Type="Constant" Value="0."/>
    <IntVar Index="2" Type="Constant" Value="0."/>
    .
    .
    <IntVar Index="107" Type="Constant" Value="0."/>
  </Material>
</Materials>
```

□ The SOTERIA umat BCC implementation :

- A set of **Fortran files** (L. Vincent)
- Compatible with the **umat** format

\$ ls *.F

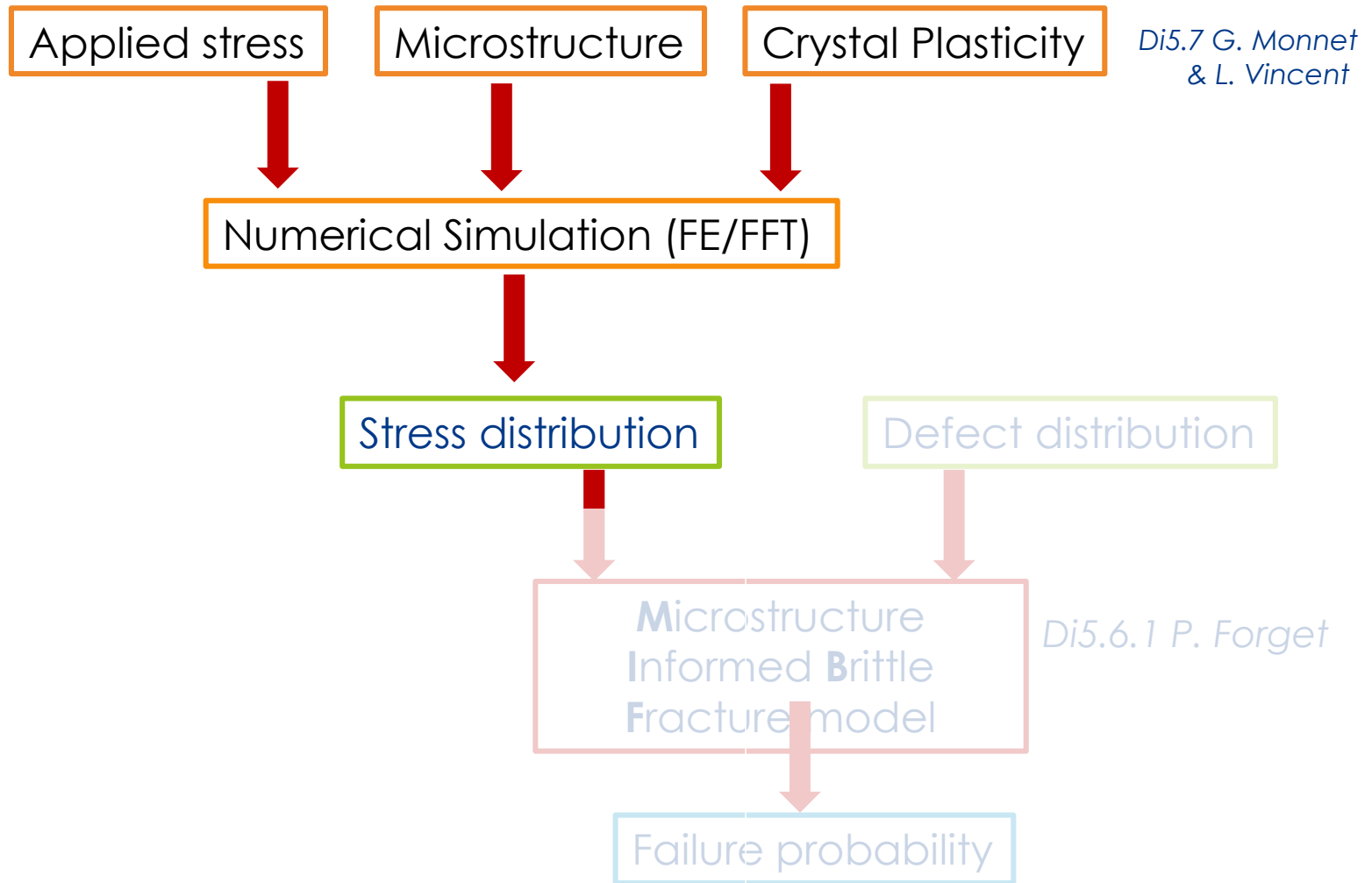
```
det.F
inv.F
jaconr86.F
ugd_algebre.F
ugd_cine.F
ugd_rk21.F
ugd_rk43.F
ugd_rkmod.F
ugd_umat.F
umat.F
```

\$ more umat.F

```
SUBROUTINE UMATBCCSOTERIA ( STRESS, STATEV, ddsdde, sse, spd, scd,
&
& rpl, dasdat, drplde, drpldt,
& STRAN, DSTRAN, TIME, DTIME,
& TEMP, DTEMP, PREDEF, DPRED,
& CMNAME, NDI, NSHR, NTENS, NSTATV,
& PROPS, NPROPS, COORDS,
& drot, pnwtdt, celent, DFGRD0, DFGRD1,
& NOEL, NPT, layer, kspt, KSTEP, KINC )
.
.
.
```

- a **Makefile** to automatically generate the dynamic library **libUmatAmitex.so** from the Fortran files
- The **same Fortan implementation** of the behavior law can be used in :
 - AMITEX_FFTP
 - CAST3M (*the CEA FEM code*)
- No need to modify the AMITEX source code to introduce a new behavior law !

□ Reliability of the Reactor Pressure Vessels



□ Reliability of the Reactor Pressure Vessels

PERFORM60 (2007-2011)

FE

- Per grain average stress

SOTERIA (2014-2018)

FFT-based simulation

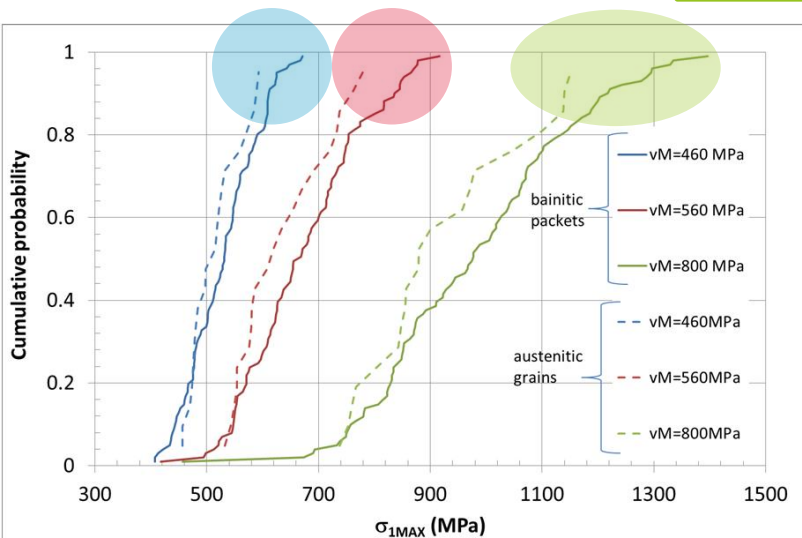
- Per grain average stress
- Large number of grains (>1000?)
- Average stress at Grain Boundaries
- Improved Crystal Plasticity law

INTRA-granular brittle fracture

Stress distribution

INTER-granular Brittle fracture

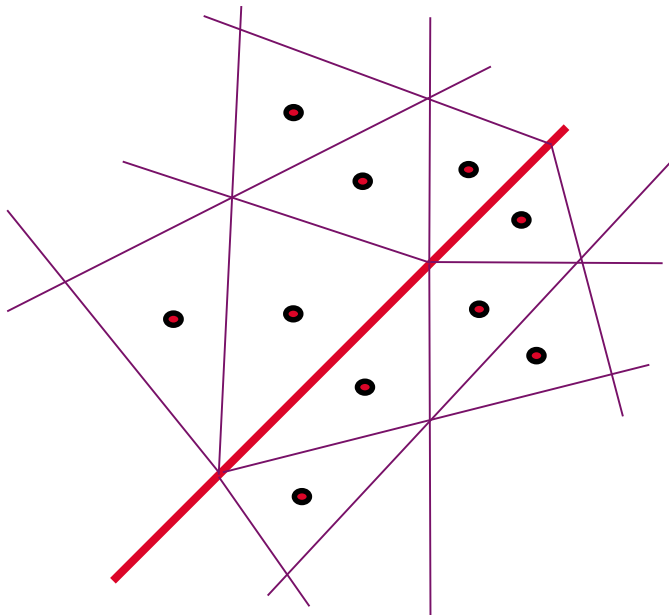
Microstructure
Informed **B**rittle
Fracture model



- ❑ Methodologies to evaluate stresses at grain boundaries from FFT simulations
- ❑ Validation with FEM (and a conforming mesh)
- ❑ Extension to Finite Strains
- ❑ Application to RPV steels

- Remark for FEM (conforming mesh)

Stresses are defined at Gauss Point (not on the grain boundary)



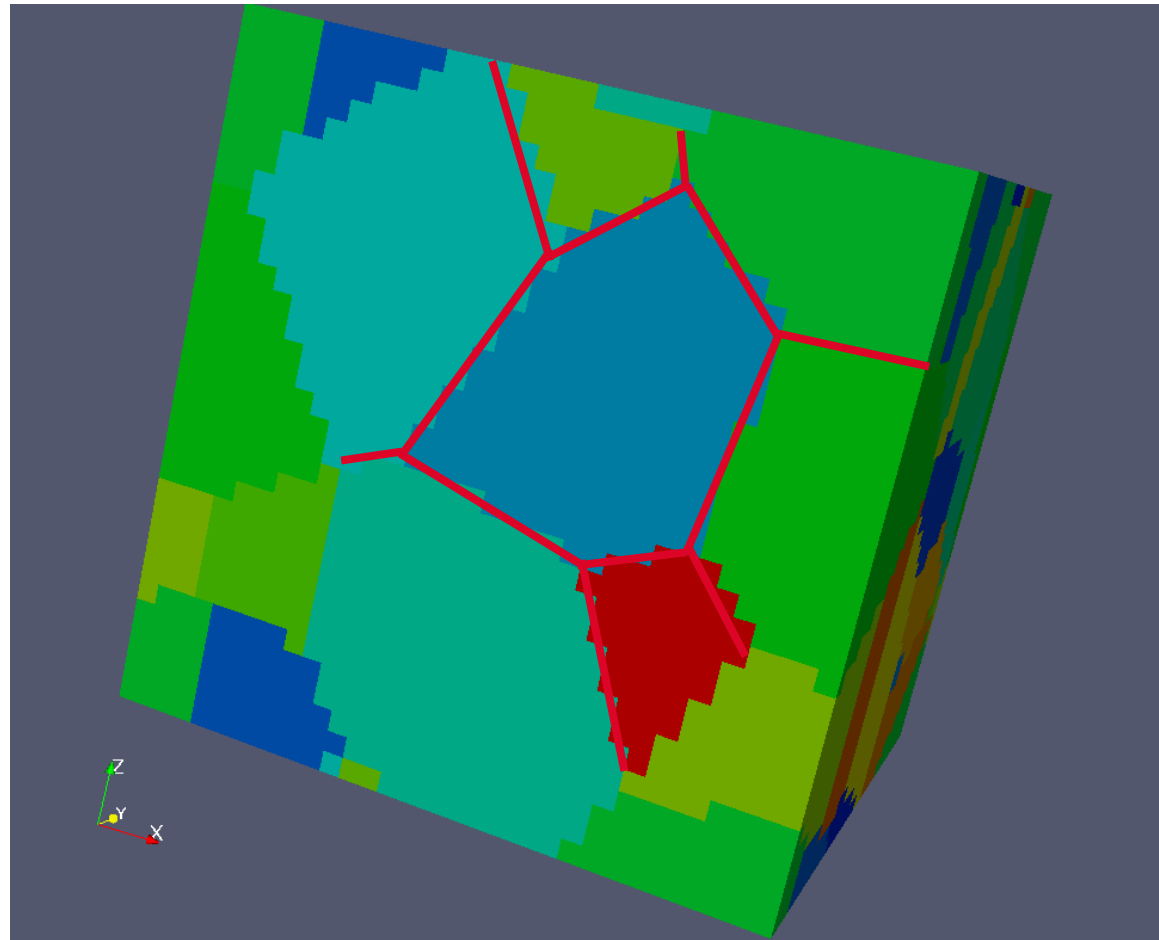
Grain Boundary (2D)

Even with a conforming mesh, the evaluation of stresses at grain boundaries **is not straightforward!**

Stresses at grain boundaries

- Methodology for FFT (regular grid)

The « mesh » does not coincide with grain boundaries

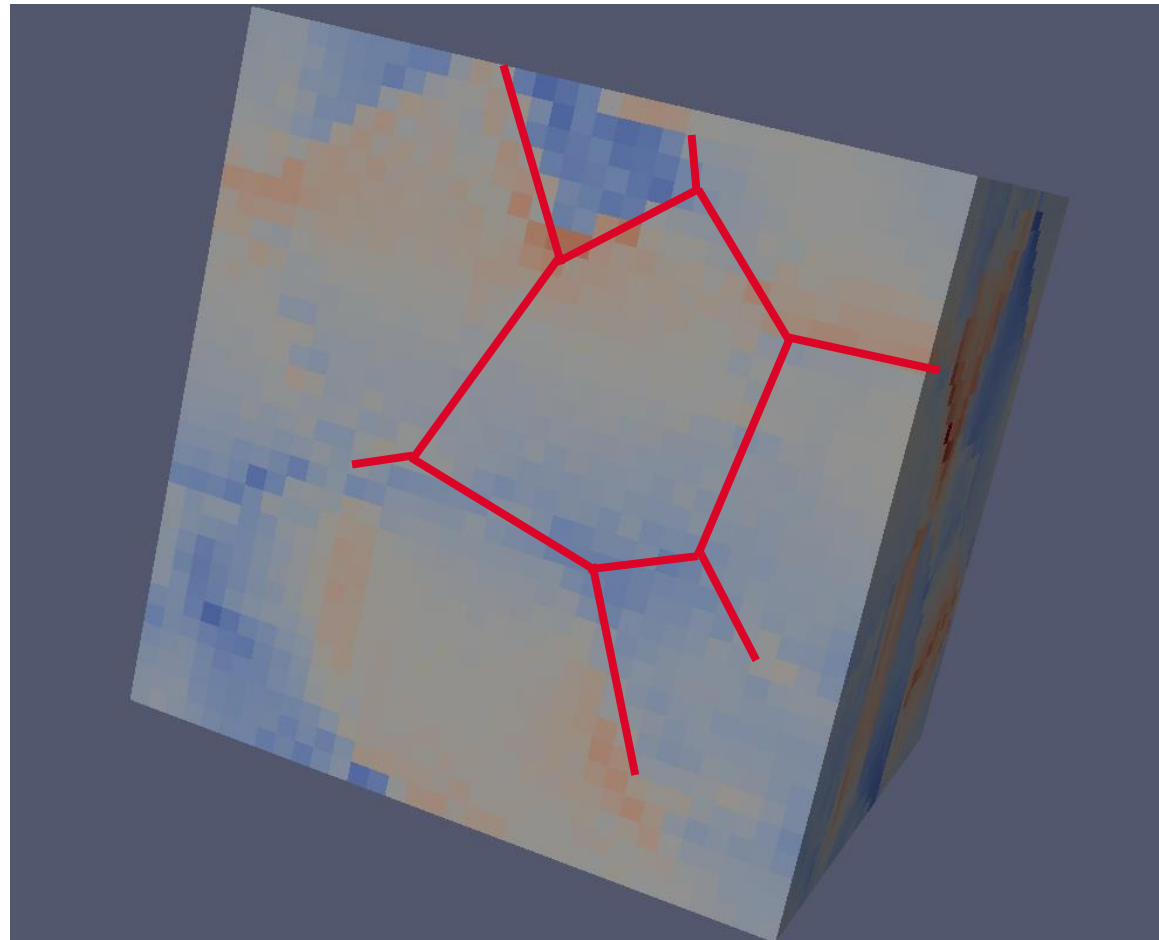


- Methodology for FFT (regular grid)

Stresses are defined on voxels

PROPOSITION 1

Post-treatment:
projection of the stress field on
grain boundaries

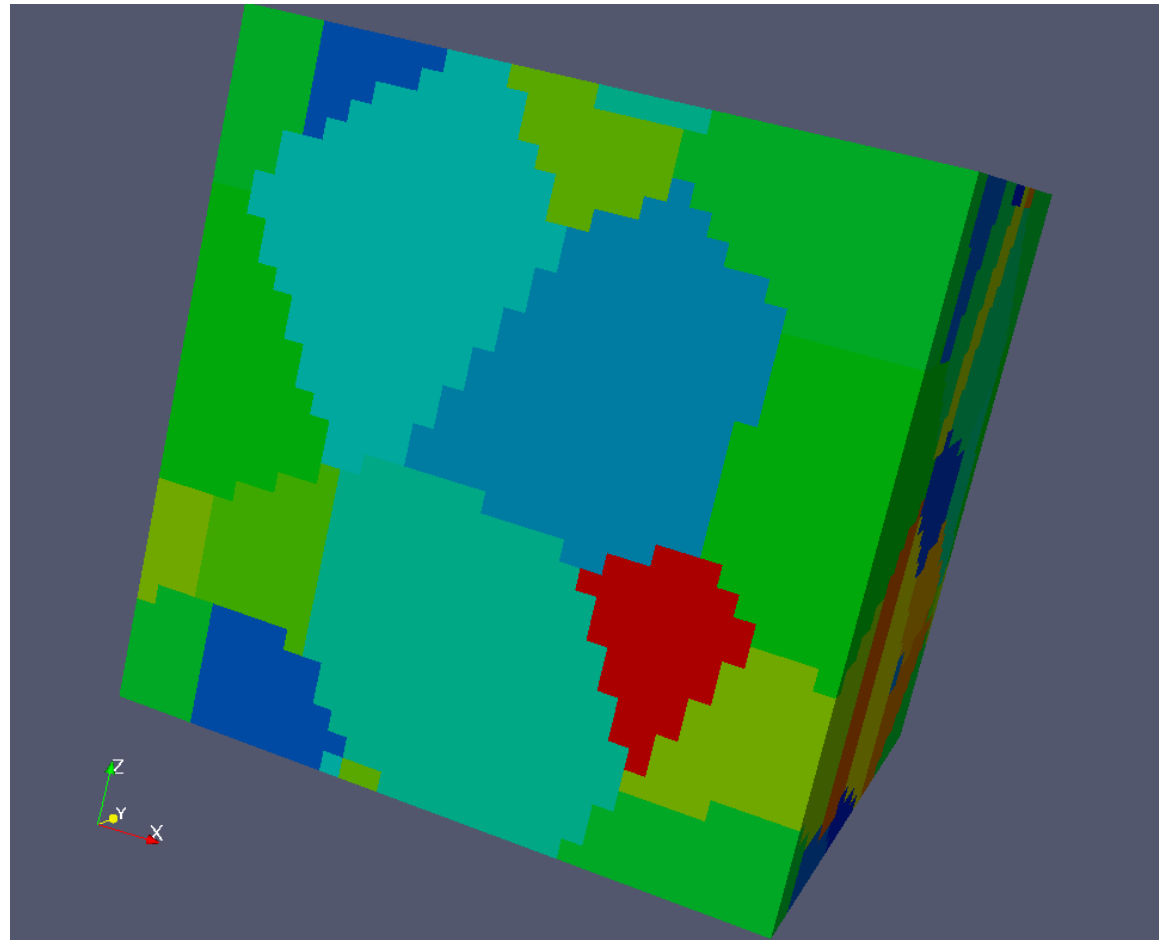


- Methodology for FFT (regular grid)

PROPOSITION 2

« Composite » voxels to account for grain boundaries in FFT simulations

+ PROPOSITION 1 (post treatment)



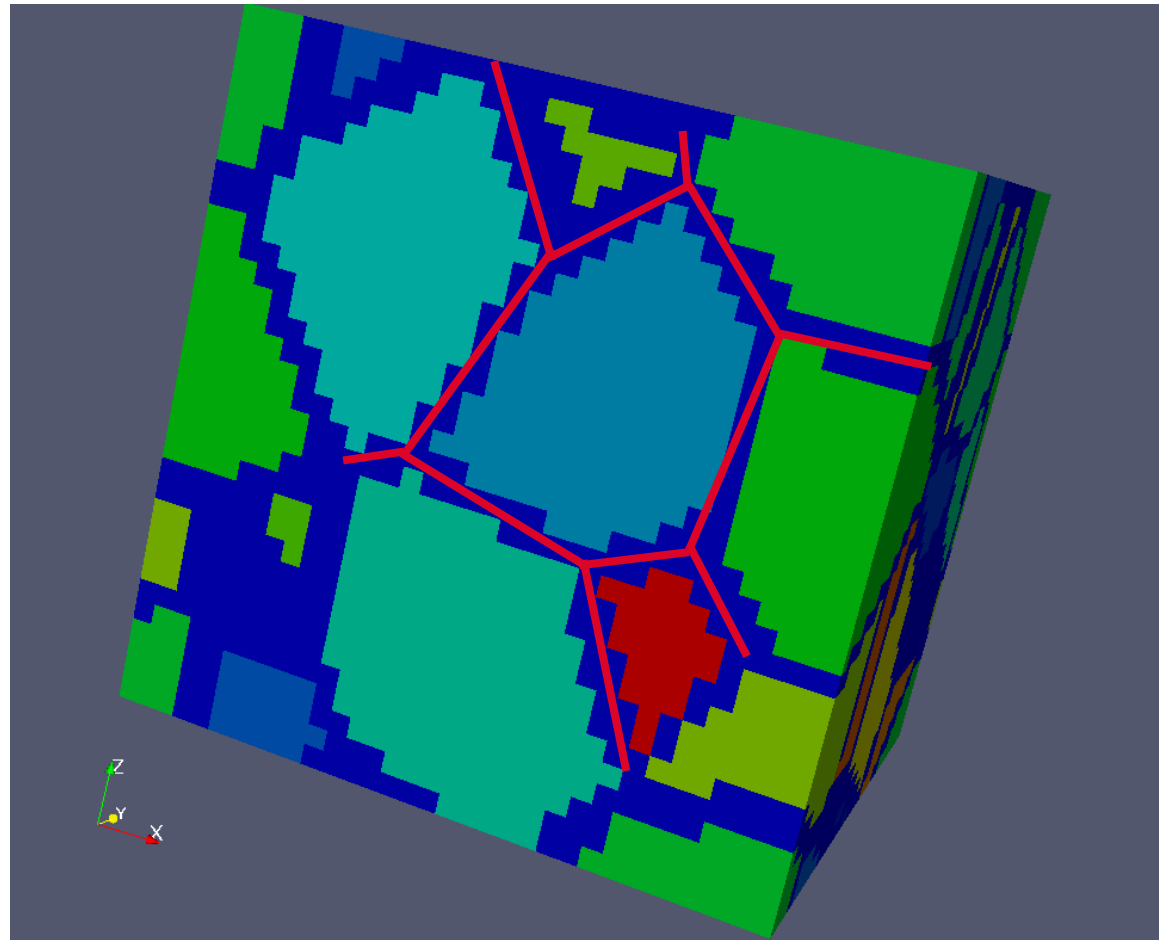
Stresses at grain boundaries

- Methodology for FFT (regular grid)

PROPOSITION 2

« Composite » voxels to account for grain boundaries in FFT simulations

+ PROPOSITION 1
(post treatment)



- Methodology for FFT (regular grid)

DEVELOPMENT OF SPECIFIC PRE AND POST TREATMENTS

✓ Grain Boundary decomposition

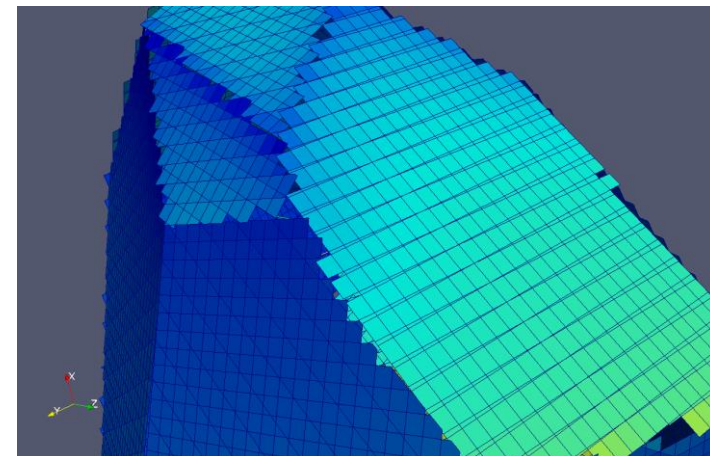
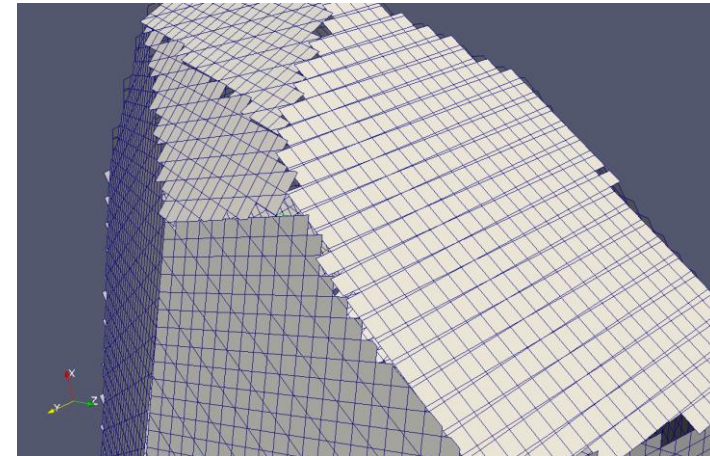
- Each **GB** is divided into « **facets** »
- One « facet » = **intersection between a GB and voxel**
 - polygon with 3, 4, 5 or 6 corners
 - *To be improved for triple lines*
- One « **composite** » voxel :
 - **Facet area** S_i
 - **Volume Fractions**
 - **Normal vector** n_i

PRE-TREATMENT

✓ Evaluating average normal stress at Grain Boundaries

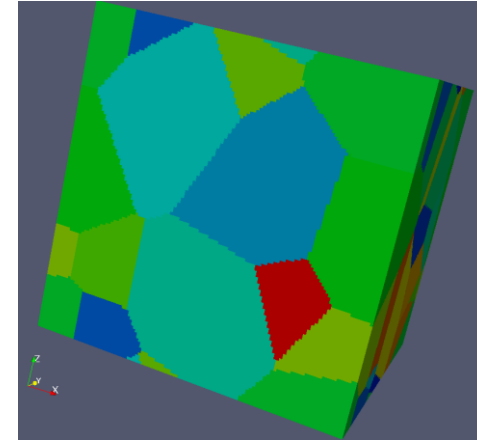
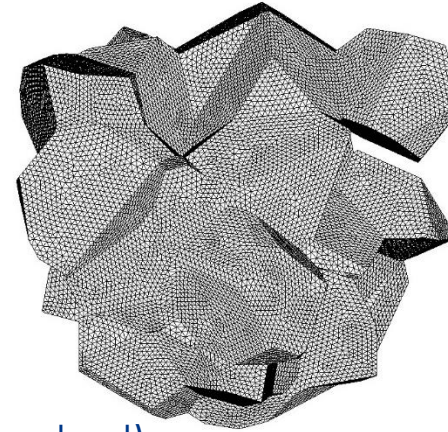
$$t = \frac{\sum n_i \cdot (\sigma_i \cdot n_i) S_i}{\sum S_i}$$

POST



□ Validation with FEM (CAST3M)

- A 27grains periodic Voronoi aggregate



- A simple Crystal Plasticity law

➔ Anisotropic elasticity (austenitic steel)

➔ **12** slip systems (FCC)

➔ Norton law ($\tau_0=200\text{MPa}$, $\mathbf{n}=10$, $\dot{\gamma}_0=10^{-4}\text{s}^{-1}$)

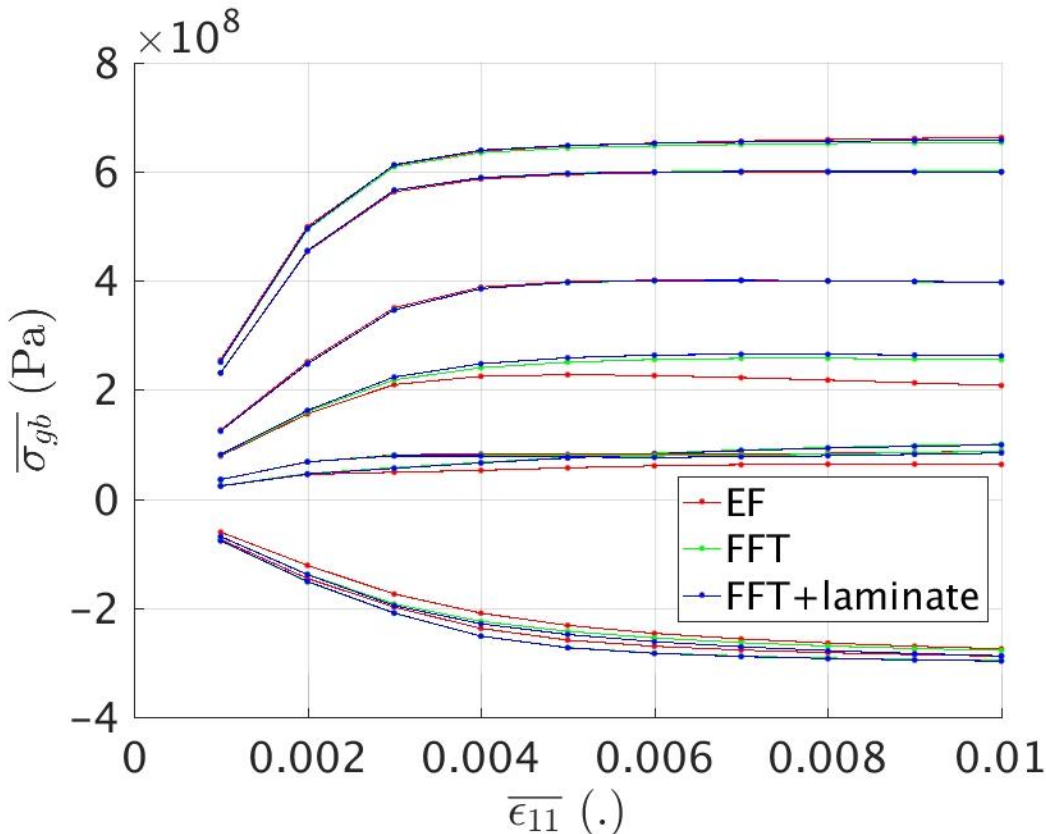
- Loading : uniaxial tensile test (1%, 10^{-4}s^{-1})

Stresses at grain boundaries



- Result :« Per Grain Boundary » average normal stresses (167 Grain Bound.)

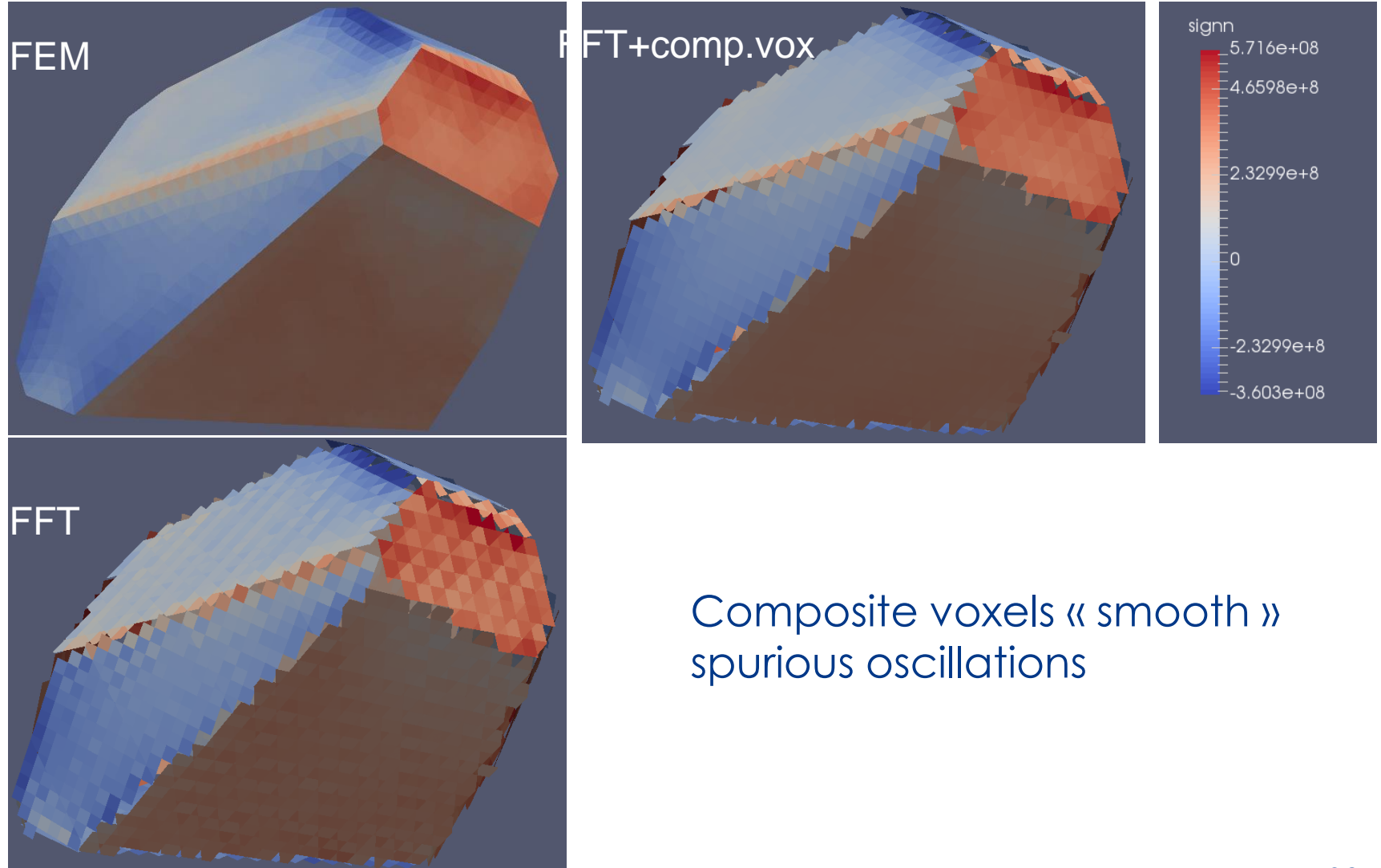
8 Grain Boundaries
4 with the best FE/FFT agreements
4 with the worst agreements



- ✓ Good agreement FFT/FEM
- ✓ FEM probably not fully converged
- ✓ No significant effect of composite voxels (FFT+laminare)

Stresses at grain boundaries

- Result : normal stress field at grain boundaries



Composite voxels « smooth »
spurious oscillations

□ Extension to Finite Strains



The grain boundary (initially plane) : rotates and deforms !

✓ **FEM** : evaluation of the normal stress on the deformed mesh of the GB

✓ **FFT** : the GB is not meshed explicitly... but

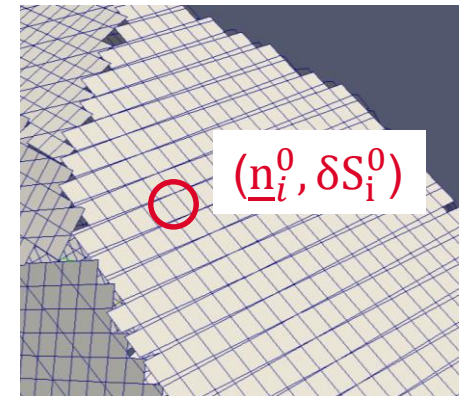
- The intersection of GB with the grid is known in the **reference configuration** ($\underline{n}_i^0, \delta S_i^0$)
- Transport equation of an infinitesimal surface vector:

$$\underline{n} dS = \det(\mathbf{F}) \mathbf{F}^{-T} \cdot \underline{n}^0 dS^0$$

$$\underline{n}_i \delta S_i = \det(\mathbf{F}_i) \mathbf{F}_i^{-T} \cdot \underline{n}_i^0 \delta S_i^0 \quad \delta S_i = \det(\mathbf{F}_i) \|\mathbf{F}_i^{-T} \cdot \underline{n}_i^0\| \delta S_i^0 \quad \underline{n}_i = \frac{\mathbf{F}_i^{-T} \cdot \underline{n}_i^0}{\|\mathbf{F}_i^{-T} \cdot \underline{n}_i^0\|}$$

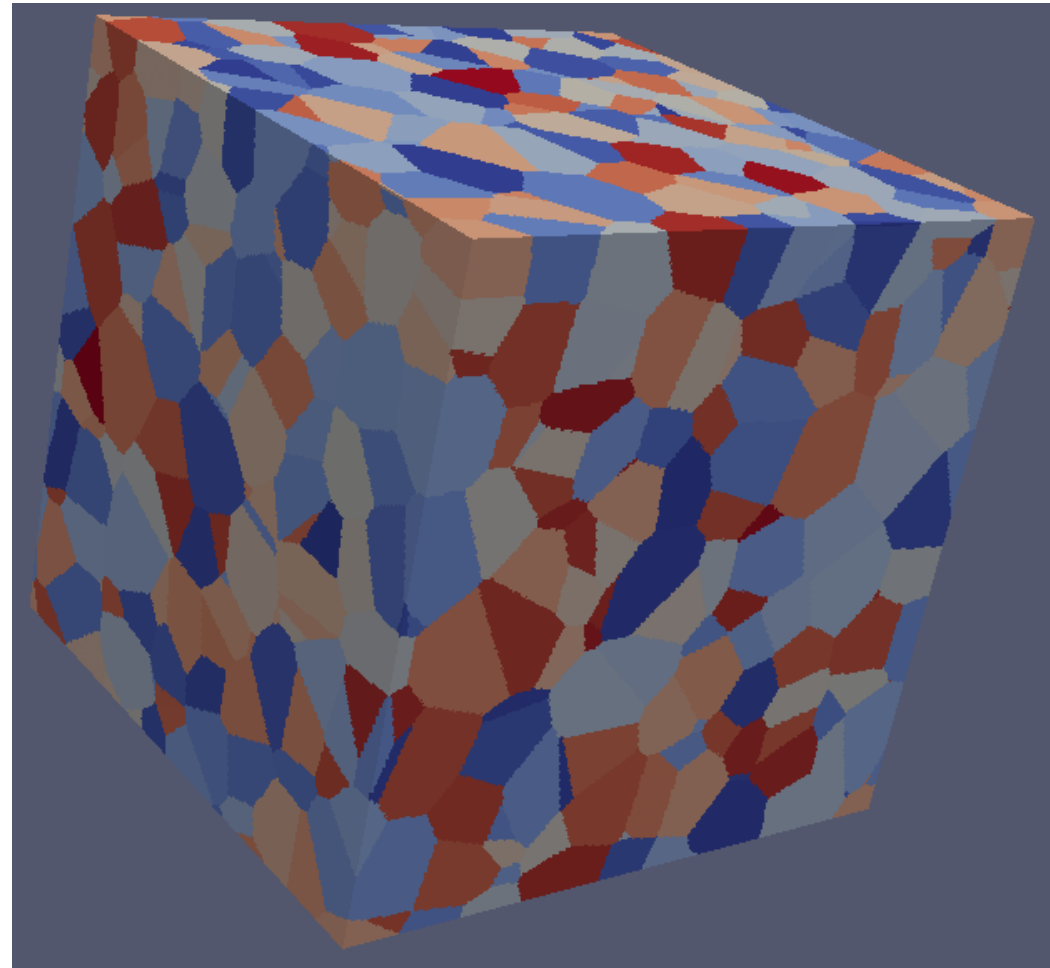
- Surface average of the normal stress in the **deformed configuration**

$$\overline{\sigma}_{gb} = \frac{\int_S \underline{n} \cdot \boldsymbol{\sigma} \cdot \underline{n} dS}{S} \cong \frac{\sum \underline{n}_i \cdot \boldsymbol{\sigma}_i \cdot \underline{n}_i \delta S_i}{\sum \delta S_i}$$

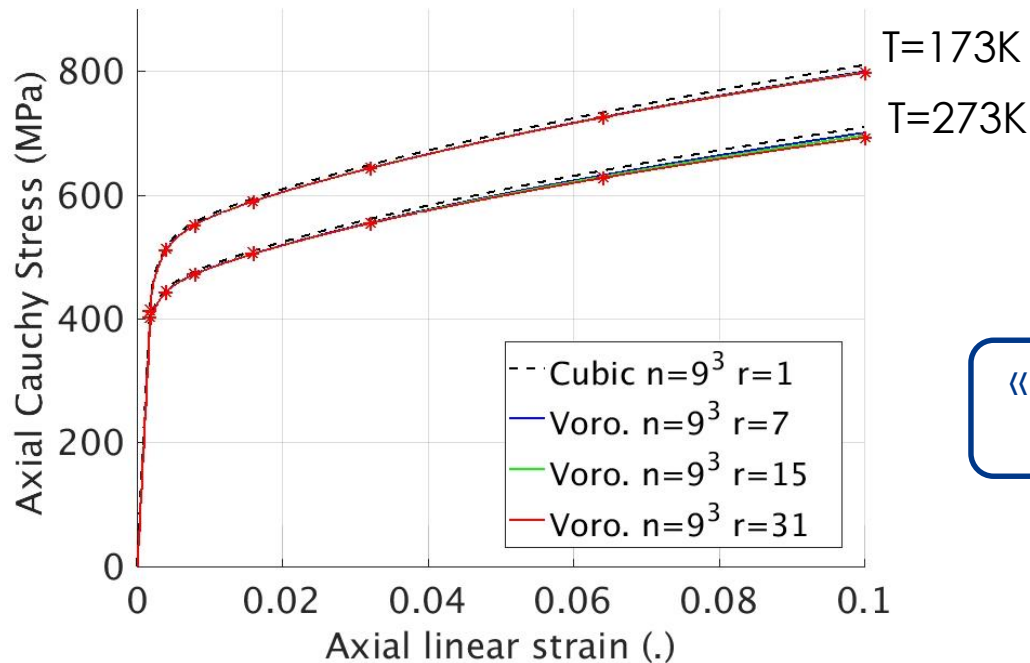


□ SOTERIA Crystal Plasticity law for BCC (Finite Strains) (G.Monnet EDF & L. Vincent CEA)

- Microstructure
 - Voronoï **729** grains
 - Resolution $r = 7, 15, 31$
- Temperatures : 173K, 273K
- Tensile test : **10%**, $3 \cdot 10^{-4} \text{ s}^{-1}$



□ Macroscopic Behavior

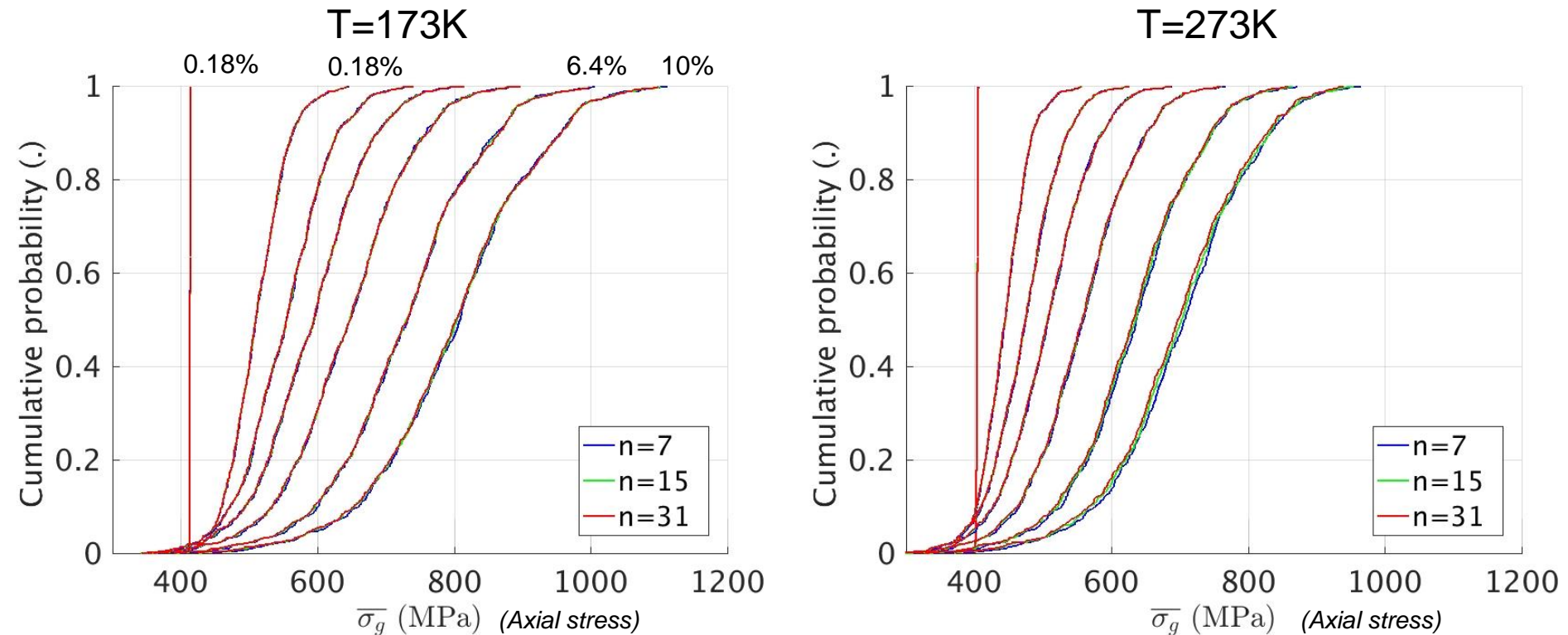


« cubic » simplified microstructure
OK for macroscopic behavior

729 grains

Cubic	r=1 :	~ 5min	/1 node (12 procs)
Voronoi	r=7 :	~30 min	/10 nodes (280procs)
	r=15:	~3h	/10 nodes (280procs)
	r=31:	~10h	/30 nodes (840procs)

□ Per Grain average stress distribution (729 grains)

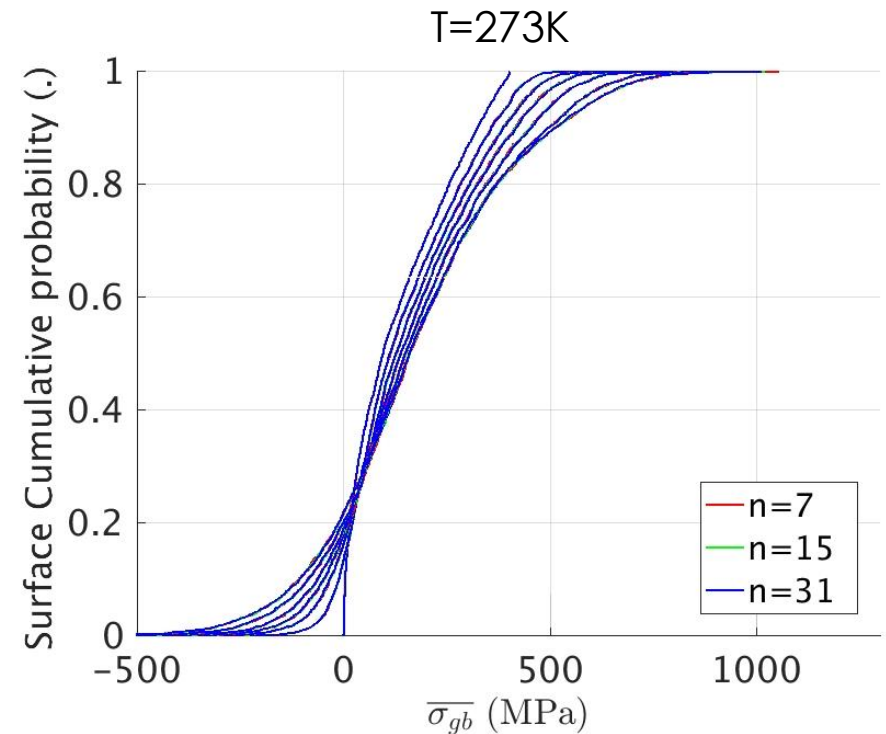
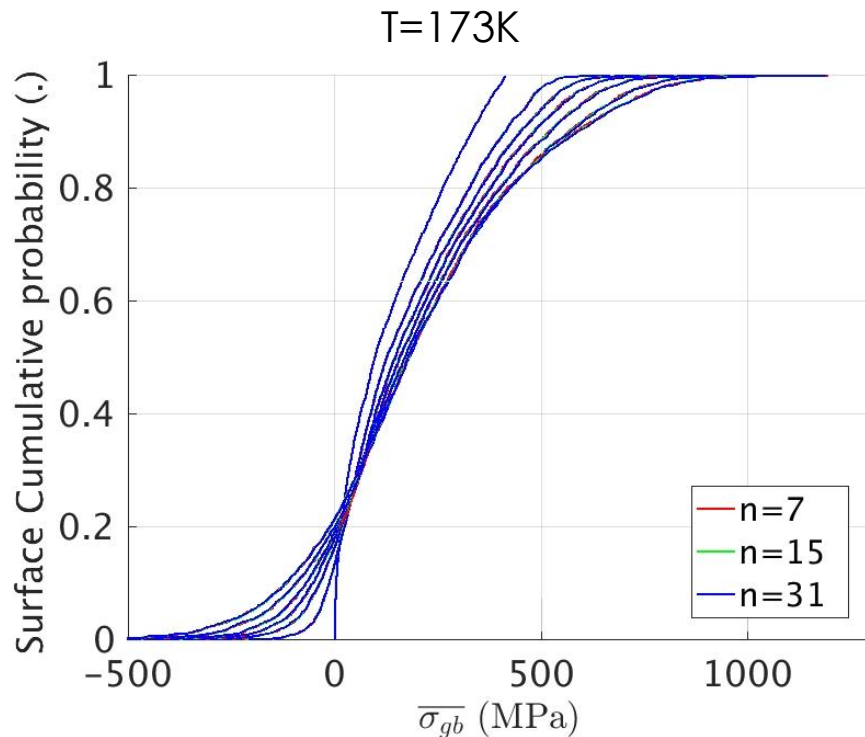


Input for MIBF for **INTRA**-granular fracture
Resolution 7 is enough !

Application to RPV

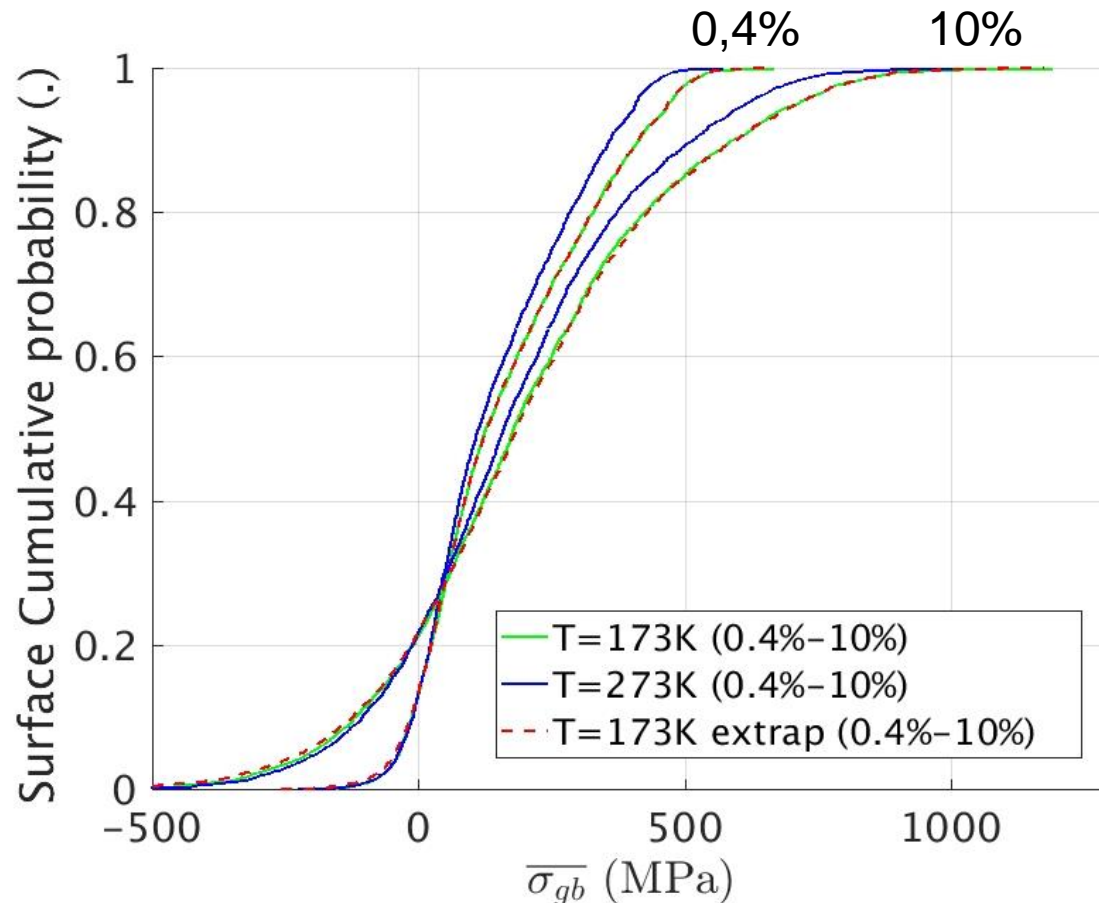


- **Per Grain Boundary** average stress distribution (4445 GB) 99% of the total area



Input for MIBF : **INTER**-granular fracture
Resolution 7 is enough !

- **Per Grain Boundary** average stress distribution (4445 GB) 99% of the total area

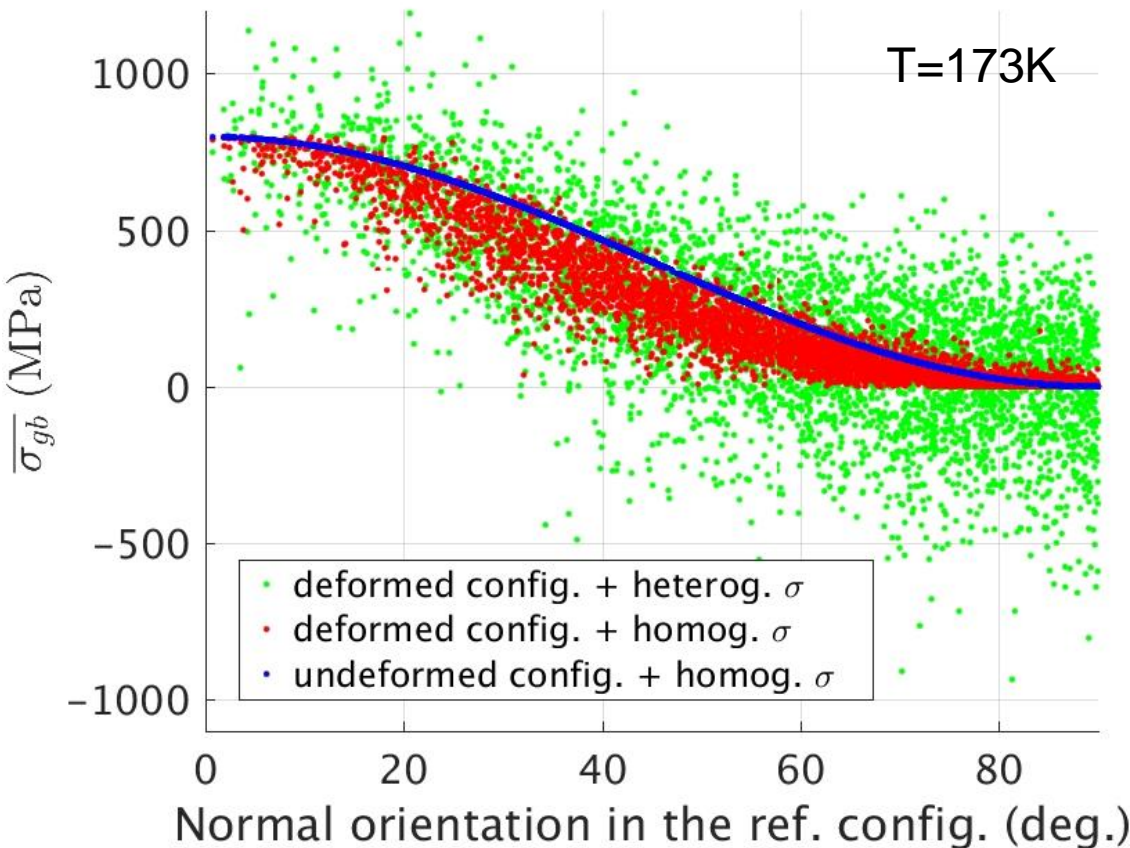


Temperature extrapolation ?

$$F_{173K}(\overline{\sigma}_{gb}) \cong F_{273K}(\overline{\sigma}_{gb} \times \alpha)$$

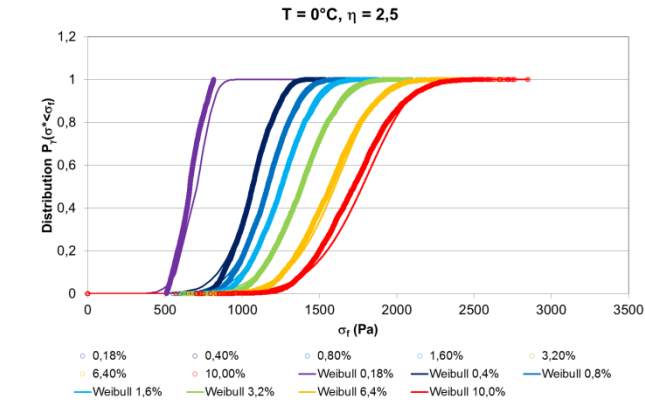
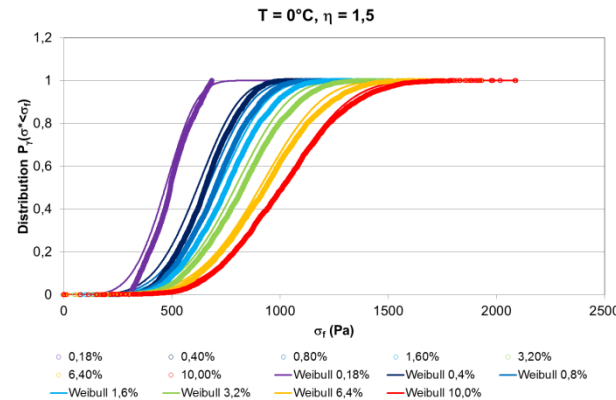
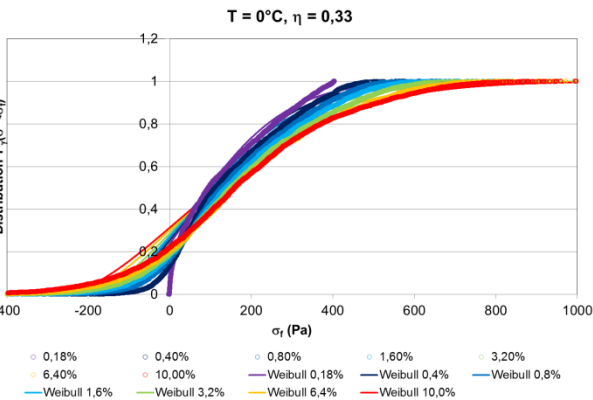
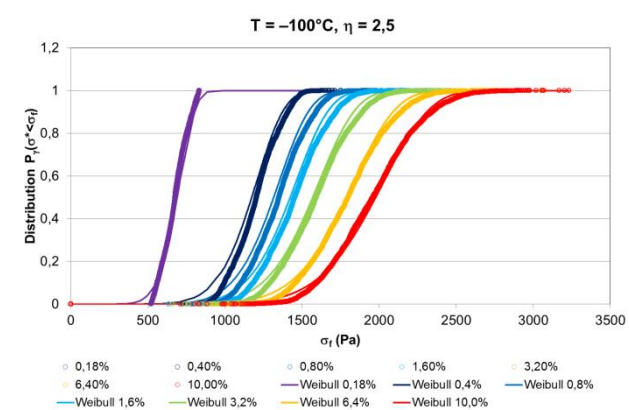
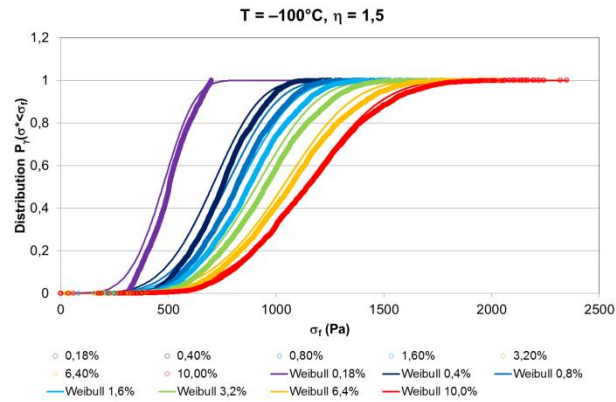
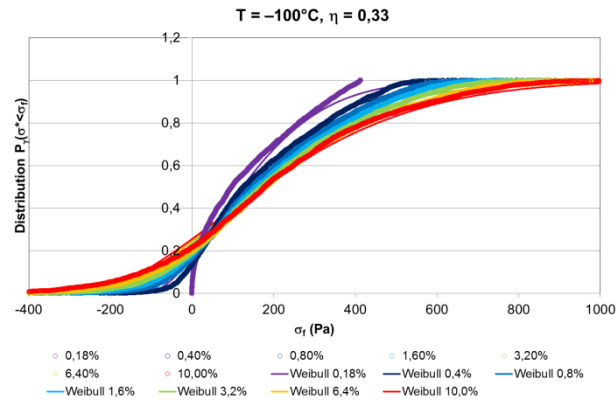
with
$$\alpha = \frac{\overline{\sigma}_{173K}}{\overline{\sigma}_{273K}}$$

- **Per Grain Boundary** average stress distribution (4445 GB) 99% of the total area
Effect of the GB rotation (and deformation)?



The rotation (and deformation) of the GBs is significant (red points)

USE for MIBF : a model fitted on the distribution of stresses at GB (P. Forget Di5.6.1)



□ GENERAL context

- FFT-based solvers :
a very powerfull technique for the simulation of heterogeneous materials
- AMITEX_FFTP :
efficient parallel code,
quite general (lots of possible applications),
quite simple to use

□ SOTERIA context

- Evaluation of stresses at GB : OK with FFT
- Application to RPV :

